

Thin Film Protective Layer with Buffering Interface

Related Application

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This is a divisional application of an application bearing serial number 09/952872 filed on Sept. 9, 2001.

Field of the Invention

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The invention relates to thin film protective layers and to methods for the deposition of thin film protective layers and more particularly to films comprising carbon and nitrogen (CNx) and even more particularly to such films as used on magnetic thin film media.

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Background of the Invention

A typical prior art head and disk system 10 is illustrated in figure 1. In operation the magnetic transducer 20 is supported by the suspension 13 as it flies above the disk 16. The magnetic transducer 20, usually called a "head" or "slider," is composed of elements that perform the task of writing magnetic transitions (the write head 23) and reading the magnetic transitions (the read head 12). The electrical signals to and from the read and write heads 12, 23 travel along conductive paths (leads) 14 which are attached to or embedded in the suspension 13. The magnetic transducer 20 is positioned over points at varying radial distances from the center of the disk 16 to read and write circular tracks (not shown). The disk 16 is attached to a spindle 18 that is driven by a spindle motor 24 to rotate the disk 16. The disk 16 comprises a substrate 26 on which a plurality of thin films 21 are deposited. The thin films 21 include ferromagnetic material in which the write head 23 records the magnetic

transitions in which information is encoded. The thin film protective layer (not shown in Figure 1) is typically the last or outermost layer.

The conventional disk 16 typically has a substrate 26 of AlMg or glass. The thin films 21 on the disk 16 typically include a chromium or chromium alloy underlayer that is deposited on the substrate 26. The magnetic layer in the thin films 21 is based on various alloys of cobalt, nickel and iron. For example, a commonly used alloy is CoPtCr. However, additional elements such as tantalum and boron are often used in the magnetic alloy.

Figure 2 illustrates one common internal structure of thin films 21 on disk 16. The protective overcoat layer 37 is used to improve wearability and corrosion. The materials and/or compositions which are optimized for one performance characteristic of an overcoat are rarely optimized for others. The most commonly used protective layer materials for commercial thin film disks have been carbon, hydrogenated carbon (CH_x), nitrogenated carbon (CN_x) and CN_xHy. Efforts to optimize overcoat properties have included use of a layer structure using different materials and/or compositions for each of two or more layers in the overcoat structure. For example, U.S. patent 5,942,317 issued to R. White describes the use of a graded CH_x protective layer wherein the hydrogen content is highest at the film's surface to take advantage of the lower polar surface energy characteristic of higher hydrogen levels (which improves corrosion resistance) and is lowest at the interface with the magnetic layer to optimize the adhesion properties. The midlevel of the CH_x film is likewise optimized by having an intermediate hydrogen concentration which has a high hardness to improve wearability. The variations in the hydrogen content can be continuous or discrete. For example, a protective layer structure with three sublayers with lower hydrogen concentration nearest the magnetic layer, intermediate hydrogen concentration in the middle sublayer and high hydrogen concentration at the surface is suggested in White '317. Hardness and density are reduced by the presence of hydrogen in certain percentage ranges; thus, the overcoat structure of White '317 is hardest and densest at the interface with the magnetic layer.

In U.S. patent 5,679,431 Chen, et al., describe the use of a bilayer protective overcoat in which the initial sublayer is carbon, titanium or chromium and the surface sublayer is CH_x or CN_x. The problem being addressed in Chen '431 is diffusion of nitrogen or hydrogen into the magnetic layer over time. The
5 initial sublayer is intended to act as a diffusion barrier.

U.S. patent 6,086,730 to Liu, et al., describes a method for sputtering a carbon protective layer with a high sp³ content which involves applying relatively high voltage pulses to the carbon target. Liu '730 asserts that the resulting carbon overcoat has good durability and corrosion resistance down to low
10 thicknesses.

In order to improve the performance of magnetic thin film media the protective overcoat 37 must be made as thin as possible to reduce the separation from the magnetic transducer 20 and the magnetic thin film 33 while maintaining the protective function.
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Summary of the Invention

The applicants disclose a method for sputtering a protective layer which
5 allows the protective layer to be ultra-thin with improved durability over prior art
films. The method reduces the kinetic energy of the impinging ions during the
initial period of deposition to form a buffering interface which reduces the
interpenetration of the atoms of the protective layer into the underlying film. The
lower energy ions form a less dense and softer film than do higher energy ions.
10 In the method of the invention the sputtering of the overcoat preferably begins
with zero (or very low) bias voltage applied to the underlying film. This "low
energy" phase of the deposition results in minimal ion implantation in the
underlying film. The "low energy" deposition continues only as long as it takes to
form a buffer layer of the overcoat material on the underlying film. The buffer
15 layer deposited in this phase is relatively soft and is, therefore, not sufficient for a
complete overcoat. The "high energy" phase of the process begins with
increases in the magnitude of the negative bias voltage applied to the underlying
film. The higher energy imparted to ions in the plasma result in a denser and
harder film being formed over the initial buffer layer. The initial buffer layer
20 reduces the interpenetration of the higher energy ions into the underlying film.
The protective layer preferably comprises carbon and nitrogen. The protective
layer structure of the invention is preferably used over a magnetic layer on thin
film magnetic media. The protective film produced by the method of the
invention has a relatively lower density at the interface with the underlying film
25 and a relatively higher density at the surface.

Brief Description of the Figures

Figure 1 is a symbolic illustration of the prior art showing the relationships between the head and associated components in a disk drive.

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Figure 2 is an illustration of a layer structure for a magnetic thin film disk according to the invention.

Figure 3 is a graph of the anticipated distribution of the depth of 50 ev carbon ion implantation into a CoPtCr magnetic film.

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Detailed Description of the Invention and the Preferred Embodiments

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Figure 2 illustrates a cross section of a magnetic thin film disk embodying the protective layer structure of the invention. The film structure illustrated contains only one magnetic layer 33 and one underlayer 31. However, the protective layer structure of the invention is not dependent on any particular underlying film structure so long as the final layer below the overcoat is conductive. The protective layer of the invention, therefore, may be used on any combination of multiple magnetic layers, underlayers and seed layers. The interface 42 between the magnetic layer 33 and the protective layer 37 is the region of the protective layer 37 that has the lowest density (indicated by the spacing of the small circles in the drawing) and the surface of the protective layer 37 has the highest density.

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The preferred material for the protective layer 37 is CN_x. Other elements such as hydrogen may be added to the film in relatively small atomic percentages. The preferred method of depositing the protective layer 37 of the invention is by sputtering using known techniques for forming a CN_x film with the exceptions noted below. In the typical process for forming a CN_x film a graphite

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target is used and nitrogen is introduced into the sputtering chamber as a gas. The relative concentration of nitrogen in the deposited film is controlled by modulating the partial pressure of the nitrogen gas in the chamber. Lower partial pressures of nitrogen result in lower concentrations of nitrogen in the film as would be expected. As is well known to those in the sputtering arts, the precise partial pressures of nitrogen and the working gas (typically argon) are derived empirically for each unique combination of equipment used in the sputtering process.

The preferred embodiment of the invention has from 5 to 25 at.% nitrogen in the protective layer. The preferred thickness of the protective layer is from 2 to 9 nanometers.

The method of the invention includes modulating the bias voltage applied to underlying film. The use of negative bias voltages applied to metallic substrates is well known. The larger the magnitude of the voltage, the more kinetic energy is imparted to the positive ions as they are accelerated toward the substrate. Higher energy ions result in a denser, harder and smoother overcoat film due, at least in part, to resputtering effects. The higher energy ions also interpenetrate the underlying film to a greater depth than do lower energy ions. This interpenetration is considered to be negligible for many applications since the depth of penetration is small in comparison to the film thickness. However, in applications such as magnetic thin film media, the films are sufficiently thin that the interpenetration of atoms into the lattice of magnetic materials is undesirable. Figure 3 is a graph of the anticipated distribution of the depth of 50 ev carbon ion implantation into a CoPtCr magnetic film. Using the method of the invention the initial bias voltage is essentially zero which reduces the average energy of the impinging ions to a few electron volts. The interpenetration of the overcoat atoms into the magnetic film is negligible at this energy level.

Moreover, for ultra-thin overcoats (for example, 0.5 to 2.5 nm) the performance of the overcoat depends critically on the nature of the interface with the underlying film. Although negative voltage bias improves the overcoat itself,

it has been found by the applicants to degrade the interface for overcoats on the order of 2.5 nm thick.

In an experiment performed by the applicants, prior art sputtering techniques using -50v bias were used to deposit 2.5 nm CNx overcoats on a batch of thin film magnetic disks. The disks were then subjected to the finishing and testing process that is normally used for large scale manufacturing of magnetic disks which includes burnishing the surface of the disks using special heads with leading edges designed to cut off the higher protrusions. To be commercially usable the overcoat on the disks must be able to withstand this burnishing and still present a surface to the slider of the magnetic transducer over which the slider can "fly" without excessive disturbance. In the experiment 85% of the prior art disks with 2.5 nm CNx overcoats failed to provide a flyable surface after burnishing, i.e., the usable yield was 15%.

A second batch of otherwise identical disks was prepared using the method of the invention to sputter 2.5 nm CNx overcoats. The particular sputtering setup required approximately four (4) seconds to deposit 2.5 nm of CNx. For the initial one (1) second, no voltage bias was applied to the underlying CoPtCr film. The underlying film was then subjected to -50v dc bias for the remainder of the deposition. This second batch of disks was then burnished and tested for flyability. These disks passed the flyability test 87% of the time representing nearly a six-fold increase in yield over the prior art disks.

In the experiment described above the bias was rapidly switched from 0 to -50v dc after the initial period in which the lower density CNx material for the buffering interface was formed. The bias can also be increased gradually, as long as the low and high density portions of the film are given adequate time to form. The preferred range of dc bias voltages for the high voltage period is from -50v to -400v.

The method of the invention can also be used with dual cathode pulsed sputtering techniques. With this technique the pulsing of opposing targets provides considerable ion bombardment of the films deposited on grounded

substrates, therefore, for this embodiment the preferred bias voltages are in the range of 0 (ground) to -200v.

Applying bias to disks with conductive substrates such as the NiP coated AlMg substrates is a straightforward process. The edges of the disk are held during sputtering by conductive material to which the bias voltage is applied. Whether the points of electrical contact are blocked or shadowed during the deposition is irrelevant since the substrate itself is conductive. However, for nonconductive substrates such as glass the bias voltage must be applied to a conductive film on the disk, so shadowing must be taken into account. There are several different types of mechanical systems used to load and support disks while they are being sputtered. Regardless of what type of system is being used, the contact points for delivery of the bias voltage to the conductive film on which the protective layer of the invention is to be formed must not have been shadowed during the deposition of the conductive film(s). This condition is satisfied if the disk is held at different points during the deposition of the overcoat other than the points at which the disk was held during the deposition of the conductive film. A small rotation of the disk after the deposition of the conductive film is sufficient to move the contact points to locations where the conductive has been adequately formed. Since the magnetic thin films in question are on the order of ten's of nanometers thick care must be taken not to overheat the thin film through which the bias current flows.

The atomic percent compositions given above are given without regard for the small amounts of contamination that invariably exist in sputtered thin films as is well known to those skilled in the art.

The invention has been described with respect to use on thin film magnetic disks, but other uses and applications which can benefit from the properties of the protective layer structure of the invention will be apparent to those skilled in the art.